Synthesis and characterization of liquid-crystalline side group polymers with benzylideneaniline as mesogenic moiety

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Two new thermotropic liquid-crystalline side group polymers were synthesized, characterized and compared with the liquid-crystalline monomeric analogues. Some packing features of these polymeric liquid crystals are discussed.

Investigations were carried out by differential calorimetry (D.S.C.), polarized microscopic observation and X-ray diffraction on non-aligned and magnetic field-aligned samples in the wide and small angle region.

The synthesized polymers contain as mesogenic moiety a benzylideneaniline group which is attached in the 4 position via a hexamethylene spacer to a polymethacrylate backbone. The benzylideneaniline group is substituted in 4' position with an ethoxy or butoxy group (PET or PBu). The monomeric analogues are denoted MEt and MBu. The two polymers show a phase sequence crystalline-smectic A-nematic-isotropic. The liquid-crystalline temperature range is observed between 90 and 150°C. The monomeric MEt exhibits only a monotropic nematic, MBu an enantiotropic nematic and a smectic A phase.

1. Introduction

Liquid-crystalline side group polymers have been widely studied for some time because these compounds should combine the favourable behaviour of liquid-crystalline phases with polymeric properties. However, many problems concerning their physical behaviour have not been solved: the influence of the mesogenic side group on the structure of liquid-crystalline phases, the prediction of phase sequences and a uniform structural model are not yet available. The proposed models depend on the method of investigation, with the placement and role of the main chain not clearly established.

The model evaluation by X-ray methods suffers from limited data even in the crystalline state, because of the poorly developed long range order for some side group polyacrylate and polymethacrylate. Comparative studies of very different systems in a systematic way might lead to progress in solving the structural problems. The synthesis of new types of side group polymers appears to be a necessity and should result in materials with various phase sequences.

In low molar mass liquid crystals the benzylideneaniline group belongs to the oldest and most widely used mesogenic moiety and shows a large variety of mesomorphy. However, this mesogenic group was not fully exploited as a side group in polymeric liquid crystals. In the beginning of the development of liquid crystal side group polymers, alkyl and alkoxy substituted benzylideneaniline was attached without a spacer to a polyacrylate and polymethacrylate backbone [1] and poor mesophase behaviour was detected. An Sm phase was obtained for carboxyl substituted benzylideneaniline bound without a spacer to a polymethacrylate chain [2]. After introduction of the spacer concept, better phase behaviour was observed. A nitrile [3] and butyl [4] substituted benzylideneaniline was fixed with spacers of 6 and 11...
methylenegroups (C<sub>6</sub> and C<sub>11</sub>) to a polyacrylate and polymethacrylate main chain and led to S<sub>A</sub> and S<sub>C</sub> phases for the nitrile-substituted mesogen and to S<sub>A</sub> and S<sub>B</sub> phases with the butyl-substituted one and a spacer length of C<sub>11</sub> for both polymers.

As a continuation of this work, the alkoxy substituted benzylideneanilines were synthesized for a comparative study of monomeric and polymeric liquid crystals. The mesogenic group was attached by a hexamethylene spacer to a polymethacrylate chain. The polarity of the alkoxy group lies between the nitrile and butyl group and might cause a different phase sequence not yet observed with this side group mesogen.

2. Experimental

New compounds (see figure 1) were synthesized according to well known methods. The reaction paths are given in figure 2 and documented in [5]. An elementary analysis was performed by the microanalytic laboratory Beller in Göttingen, and the results are listed in Table 1.

![Figure 1](image-url)

Figure 1. Monomeric and polymeric compounds synthesized, abbreviations with regard to the terminal group of the mesogenic moiety (M, monomer; P, polymer) and numbering.

<table>
<thead>
<tr>
<th>Compound</th>
<th>C/per cent</th>
<th>H/per cent</th>
<th>N/per cent</th>
</tr>
</thead>
<tbody>
<tr>
<td>MEl</td>
<td>73·38 (73·32)</td>
<td>7·67 (7·63)</td>
<td>3·55 (3·42)</td>
</tr>
<tr>
<td>MBu</td>
<td>73·36 (74·11)</td>
<td>8·10 (8·06)</td>
<td>3·25 (3·20)</td>
</tr>
<tr>
<td>PET</td>
<td>73·27 (73·32)</td>
<td>7·55 (7·63)</td>
<td>3·40 (3·43)</td>
</tr>
<tr>
<td>PBu</td>
<td>73·37 (74·11)</td>
<td>8·17 (8·06)</td>
<td>3·29 (3·20)</td>
</tr>
</tbody>
</table>

Polarization microscopic investigations were carried out with an Olympus BH-2 microscope equipped with a hot stage of Mettler FP 800. Since characteristic textures were obtained more easily on cooling from the isotropic melt, the samples were melted on a glass plate and covered with a second glass plate, and air bubbles were removed through a light pressure on the cover slide.

Nickel-filtered Cu-K<sub>α</sub> radiation was used for X-ray studies in a vacuum flat film and Kratky compact camera, both equipped with electrically heated sample holders. A magnetic field perpendicular to the X-ray beam was available in the flat film camera [6]. Two temperature resistant permanent magnets at a distance of 1·5 mm created a
magnetic flux of 0.9 T. The sample was placed in a Mark capillary and X-ray exposures were taken upon cooling from the isotropic melt for the samples in the mesophases. The Kratky camera with a position-sensitive counter (Firma Braun, Munich, F.R. Germany) served for studies of d-spacings in the small angle region of non-aligned samples. In contrast with the flat film exposures, the measurements were performed upon heating. The thermal investigations were carried out with a Perkin-Elmer D.S.C.-7 apparatus and the onset and peak temperatures were taken, listed and discussed.

3. Results and discussion

3.1. Monomers

The monomers MEt (1b) and MBu (1c) (cf. figure 1) differ little in their chemical constitution but show quite different phase behaviour. Only a monotropic, nematic mesophase is detected for MEt and appears between 85 and 74 °C in the polarization microscope, mostly as pseudo-isotropic phase, upon cooling. Observations in the conoscopic mode reveal a uniaxial, optically positive structure. Occasionally a Schlieren-texture was obtained as represented in figure 3, which was difficult to reproduce.

The X-ray pattern (see figure 4) of a magnetic field-aligned nematic sample of MEt shows, besides an equatorial halo with \( d = 4.5 \text{ Å} \), a weak and diffuse meridional reflection at \( d = 28.0 \text{ Å} \), which can be clearly distinguished from the sharp layer reflections of smectic phases. The third order of this reflection with \( d = 9.3 \text{ Å} \) is still observable.
Figure 3. Nematic Schlieren-texture of MEt observed between crossed polars in the microscope (× 100) at 78·1°C.

Figure 6. Fan-shaped texture of smectic A phase for MBu observed between crossed polars in the microscope (× 270) at 76·3°C.
Figure 4. X-ray pattern of the nematic sample of MEt oriented in a magnetic field at 80°C. The magnetic field direction is perpendicular to the X-ray beam, and is placed horizontal in this representation. It also represents the fibre axis of this X-ray exposure.

The D.S.C. measurements confirm the monotropy of the phase behaviour of MEt (see figure 5). It is noteworthy that the isotropic–nematic transition at 79.7°C in D.S.C. measurements is about 5°C lower than by polarization microscopic observations. The sample seems to be very sensitive to the preparation methods. The phase transition temperature was determined for a homeotropic oriented sample in the polarization microscope and for an unoriented sample in the D.S.C. apparatus, and therefore should not be compared.

Table 2. Phase behaviour of the compounds investigated derived from polarization microscopic observation (temperatures listed in °C).

<table>
<thead>
<tr>
<th>Compound</th>
<th>K</th>
<th>S_A</th>
<th>N</th>
<th>I</th>
</tr>
</thead>
<tbody>
<tr>
<td>ME</td>
<td>Heating</td>
<td>82-85</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td></td>
<td>Cooling</td>
<td>74.1</td>
<td>-</td>
<td></td>
</tr>
<tr>
<td>MBu</td>
<td>Heating</td>
<td>74.3</td>
<td>78.5</td>
<td>82.2</td>
</tr>
<tr>
<td></td>
<td>Cooling</td>
<td>55.9</td>
<td>78.2</td>
<td>82.2</td>
</tr>
<tr>
<td>PEt</td>
<td>Heating</td>
<td>88.2</td>
<td>127.6</td>
<td>151.7</td>
</tr>
<tr>
<td></td>
<td>Cooling</td>
<td>90.1</td>
<td>127.2</td>
<td>150.6</td>
</tr>
<tr>
<td>PBu</td>
<td>Heating</td>
<td>97.8</td>
<td>130.9</td>
<td>136.8</td>
</tr>
<tr>
<td></td>
<td>Cooling</td>
<td>96.0</td>
<td>132.3</td>
<td>137.2</td>
</tr>
</tbody>
</table>

The monomeric compound MBu shows a more complex phase behaviour than MEt (see table 2). A nematic and S_A mesophase are observed besides several crystalline modifications. The nematic phase appears as a homeotropic, uniaxial and
Figure 5. Normalized D.S.C. curves of the monomeric samples of MEt and MBu. 1, MEt upon heating at 5°C min⁻¹; 2, MBu upon heating at 5°C min⁻¹; 3, MBu upon cooling at 5°C min⁻¹; 4, MEt upon cooling at 5°C min⁻¹.

Table 3. Temperature and enthalpy of phase transitions of the monomers MEt and MBu from D.S.C. measurements.

<table>
<thead>
<tr>
<th>Compound</th>
<th>( \theta_{\text{onset}} ) /°C</th>
<th>( \theta_{\text{peak}} ) /°C</th>
<th>( \Delta H ) /J g⁻¹</th>
<th>Remarks</th>
</tr>
</thead>
<tbody>
<tr>
<td>MEt</td>
<td>77·3</td>
<td>81·1</td>
<td>116·7</td>
<td>K ( \rightarrow ) I, upon heating (5°C min⁻¹)</td>
</tr>
<tr>
<td>MEt</td>
<td>79·6</td>
<td>79·3</td>
<td>-2·1</td>
<td>I ( \rightarrow ) N</td>
</tr>
<tr>
<td>MEt</td>
<td>61·3</td>
<td>58·1</td>
<td>-100·6</td>
<td>N ( \rightarrow ) K</td>
</tr>
<tr>
<td>MBu</td>
<td>69·5</td>
<td>70·7</td>
<td>62·9</td>
<td>K₁ ( \rightarrow ) K₂</td>
</tr>
<tr>
<td>MBu</td>
<td>71·7</td>
<td>73·7</td>
<td>4·7</td>
<td>K₂ ( \rightarrow ) K₃</td>
</tr>
<tr>
<td>MBu</td>
<td>76·0</td>
<td>76·7</td>
<td>3·9</td>
<td>Sₐ ( \rightarrow ) N</td>
</tr>
<tr>
<td>MBu</td>
<td>77·6</td>
<td>78·2</td>
<td>1·8</td>
<td>N ( \rightarrow ) I</td>
</tr>
<tr>
<td>MBu</td>
<td>81·7</td>
<td>82·2</td>
<td>-1·9</td>
<td>I ( \rightarrow ) N</td>
</tr>
<tr>
<td>MBu</td>
<td>81·6</td>
<td>81·2</td>
<td>-3·9</td>
<td>N ( \rightarrow ) Sₐ</td>
</tr>
<tr>
<td>MBu</td>
<td>56·5</td>
<td>56·0</td>
<td>-3·4</td>
<td>Sₐ ( \rightarrow ) K₁</td>
</tr>
<tr>
<td>MBu</td>
<td>53·3</td>
<td>52·4</td>
<td>-50·0</td>
<td>K₁ ( \rightarrow ) K₂</td>
</tr>
<tr>
<td>MBu</td>
<td>48·5</td>
<td>45·0</td>
<td></td>
<td>K₂ ( \rightarrow ) K₃</td>
</tr>
</tbody>
</table>

optically positive structure, when prepared between two cover slides, as described for MEt. This phase shows the same X-ray pattern characteristic for the MEt compound. The Sₐ phase exhibits a focal-conic fan-like texture in the polarization microscope (see figure 6).
The X-ray pattern of a magnetic field-aligned smectic sample shows the characteristic features of an $S_A$ phase (see figure 7): a diffuse equatorial halo and sharp meridional layer reflections with $d = 30.5\,\text{Å}$, and small second and third order reflections thereof.

The significant difference between the diffuse meridional reflections of the nematic phase and the sharp smectic layer reflections of the $S_A$ phase is best demonstrated in a Kratky diagram (see figures 8 and 9).

Further phase transitions are detected by D.S.C. below the $S_A$ phase (figure 5), mostly crystalline-crystalline transitions. The $S_A$ neighbouring phase, termed $K$, in table 3, and whose Kratky X-ray diagram is represented in figure 10, shows a texture similar to a highly ordered smectic phase (see figure 11). The Kratky plot looks very simple, with sharp reflections of $d = 26.55\,\text{Å}$, well below the size of the extended molecule, and some higher orders which agree with a tilted smectic phase. X-ray exposures of oriented samples have not yet been obtained, and since no macroscopic mobility of the sample was observed the phase was denoted as crystalline.

The two monomeric compounds, MEt and MBu, are difficult to interpret with regard to thermal analysis (see figure 5 and table 3). Not only do the transition temperatures determined by D.S.C. differ from those observed in the polarization microscope, but differences are also encountered for corresponding enthalpies upon heating and cooling for the same phase transition.

3.2. Polymers

The two polymeric compounds, PEt and PBu, lead to the same sequence of phases, in contrast with their corresponding monomeric analogues (see table 2). The nematic
Figure 8. Kratky plot of the non-aligned nematic phase of MBu at 82°C (counting time 300 s).

Figure 9. Kratky plot of the non-aligned smectic A phase of MBu at 79°C (counting time 300 s).

Figure 10. Kratky plot of the non-aligned probable crystalline phase of MBu at 75°C (counting time 300 s).
Figure 11. Probable crystalline texture of MBu observed between crossed polars in the microscope (×200) at 52°C.

Figure 12. Nematic Schlieren-texture of PEt observed between crossed polars in the microscope (×160) at 135°C.
phase appears sometimes as Schlieren-texture (see figure 12), sometimes as homeotropic texture. The X-ray pattern of a magnetic field-aligned sample scarcely shows any orientation of the halo and the reflection are missing in the small angle region (figure 13).

The $S_A$ phase of PBu appears as focal-conic fan texture, and was obtained from the homeotropic nematic phase upon cooling (see figure 14). The $S_A$ phase of PET, which originates from a nematic Schlieren-texture, shows a fan texture which is not fully developed (see figure 15). A clearly noticed change of texture takes place at the nematic-$S_A$ phase transition. The X-ray patterns of oriented smectic samples show the features of $S_A$ phases (see figure 16(a)). The meridional reflections are indicative of the layer thickness. They are sharp, and higher order reflections are present (see table 4). The halo, representing the packing of the side groups, lies centred on the equator.

Upon further cooling the texture remains unchanged when the sample becomes crystalline. The X-ray pattern of the crystalline phase resembles, with regard to position of reflections, i.e. the $d$-spacings, those of the corresponding $S_A$ phases (see figures 16(a)–(d)). Only the diffuse equatorial reflections (haloes) in the smectic phase sharpen considerably in the crystalline state. These observations can be interpreted as an improvement of packing of the side groups without any changes in the thickness of the smectic layers.

D.S.C. measurements show, as well as polarization microscopic observations, a lesser supercooling of the polymers than for the corresponding monomers (see figure 17 and table 5), and the values for the transition enthalpies upon heating and cooling are in better agreement.
Synt hesis and characterization of side group polymers

Figure 14. Fan-like textures of smectic A phase for PBu observed between crossed polars in the microscope (x 420) at 132°C.

Figure 15. Fan-like texture, not fully developed, of smectic A phase for PET observed between crossed polars in the microscope (x 270) at 122.6°C.
Figure 16. X-ray pattern of different polymeric samples all oriented in a magnetic field. The magnetic field direction is perpendicular to the X-ray beam, and is placed horizontal in this representation. The fibre axes correspond to magnetic field direction. (a) Smectic A phase of PET at 105°C, (b) smectic A phase of PBu at 110°C, (c) crystalline state of PET at 25°C and (d) crystalline state of PBu at 25°C.

Table 4. d-Spacings of X-ray exposures of magnetic field-oriented samples. The reflections $d_1$, $d_2$ and $d_3$ are placed on the meridian, the halo on the equator.

<table>
<thead>
<tr>
<th>Compound</th>
<th>$\theta/°C$</th>
<th>Phase</th>
<th>$d_1$/Å</th>
<th>$d_2$/Å</th>
<th>$d_3$/Å</th>
<th>$d_{\text{halo}}$/Å</th>
</tr>
</thead>
<tbody>
<tr>
<td>MEt</td>
<td>80</td>
<td>N</td>
<td>28·0</td>
<td>-</td>
<td>9·3</td>
<td>4·5</td>
</tr>
<tr>
<td>MBu</td>
<td>77</td>
<td>N</td>
<td>29·5</td>
<td>-</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>MBu</td>
<td>76</td>
<td>$S_A$</td>
<td>30·5</td>
<td>15·1</td>
<td>9·9</td>
<td>4·4</td>
</tr>
<tr>
<td>PET</td>
<td>145</td>
<td>N</td>
<td>-</td>
<td>-</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>PET</td>
<td>105</td>
<td>$S_A$</td>
<td>27·5</td>
<td>13·8</td>
<td>9·1</td>
<td>4·4</td>
</tr>
<tr>
<td>PET</td>
<td>25</td>
<td>K</td>
<td>27·7</td>
<td>13·1</td>
<td>9·1</td>
<td>4·3</td>
</tr>
<tr>
<td>PBu</td>
<td>131</td>
<td>N</td>
<td>-</td>
<td>-</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>PBu</td>
<td>110</td>
<td>$S_A$</td>
<td>29·7</td>
<td>15·0</td>
<td>9·8</td>
<td>4·4</td>
</tr>
<tr>
<td>PBu</td>
<td>25</td>
<td>K</td>
<td>29·7</td>
<td>15·0</td>
<td>9·8</td>
<td>4·4</td>
</tr>
</tbody>
</table>
Table 5. Temperature and enthalpy of the phase transitions of the polymers PEt and PBu from D.S.C. measurements.

<table>
<thead>
<tr>
<th>Compound</th>
<th>(T_{\text{onset}}/\degree\text{C})</th>
<th>(T_{\text{peak}}/\degree\text{C})</th>
<th>(\Delta H/\text{J g}^{-1})</th>
<th>Remarks</th>
</tr>
</thead>
<tbody>
<tr>
<td>PEt</td>
<td>87.7</td>
<td>90.0</td>
<td>8.1</td>
<td>(K \rightarrow S_A) upon heating (20\degree\text{C min}^{-1})</td>
</tr>
<tr>
<td>PEt</td>
<td>127.3</td>
<td>129.6</td>
<td>1.0</td>
<td>(S_A \rightarrow N)</td>
</tr>
<tr>
<td>PEt</td>
<td>146.0</td>
<td>148.0</td>
<td>3.6</td>
<td>(N \rightarrow I)</td>
</tr>
<tr>
<td>PEt</td>
<td>144.0</td>
<td>142.1</td>
<td>-3.8</td>
<td>(I \rightarrow N)</td>
</tr>
<tr>
<td>PEt</td>
<td>126.4</td>
<td>124.1</td>
<td>-1.2</td>
<td>(N \rightarrow S_A) upon cooling (20\degree\text{C min}^{-1})</td>
</tr>
<tr>
<td>PEt</td>
<td>88.1</td>
<td>82.5</td>
<td>-8.0</td>
<td>(S_A \rightarrow K)</td>
</tr>
<tr>
<td>PBu</td>
<td>91.6</td>
<td>94.3</td>
<td>3.7</td>
<td>(K \rightarrow S_A)</td>
</tr>
<tr>
<td>PBu</td>
<td>127.4</td>
<td>130.5</td>
<td>8.6</td>
<td>(S_A \rightarrow N) upon heating (5\degree\text{C min}^{-1})</td>
</tr>
<tr>
<td>PBu</td>
<td>133.1</td>
<td>135.1</td>
<td>-7.7</td>
<td>(N \rightarrow I)</td>
</tr>
<tr>
<td>PBu</td>
<td>134.9</td>
<td>133.7</td>
<td>-3.8</td>
<td>(I \rightarrow N)</td>
</tr>
<tr>
<td>PBu</td>
<td>131.0</td>
<td>129.1</td>
<td></td>
<td>(N \rightarrow S_A) upon cooling (5\degree\text{C min}^{-1})</td>
</tr>
<tr>
<td>PBu</td>
<td>96.0</td>
<td>93.9</td>
<td></td>
<td>(S_A \rightarrow K)</td>
</tr>
</tbody>
</table>

3.3. Phase behaviour

The results for the two slightly different monomeric benzylideneaniline compounds in comparison with the corresponding side group polymethacrylate can be summarized as follows.

(a) The two polymers investigated show the same sequence of phases in contrast to the monomeric compounds, which exhibit a different phase sequence.
(b) The temperature intervals for the existence of liquid-crystalline phases is much broader for the polymers.
(c) The polymer PEt shows an $S_A$ phase which does not exist for the monomer MEt.

The polymer main chain, to which the side groups are attached, strongly influences the mesomorphy. New phases are formed and the liquid-crystalline temperature region are widened and stabilized.

It is interesting to compare the various transition enthalpies (see tables 3 and 5). The enthalpies of crystallization and melting for the polymers only amount to one-tenth of those of the monomers, all the other transitions showing comparable enthalpies in order of magnitude. This indicates that the formation of liquid-crystalline polymeric phases are primarily related by packing of the side groups, but that crystallization is mainly influenced or hindered by the main chains.

3.4. Structure

The $d$-spacing for the thickness of smectic A layers as determined by X-ray experiments of the monomeric compounds, MBu and MEt, agrees very well with that of the corresponding $d$-spacings for the layers of the polymers, PBu and PEt (see table 4). The difference in thickness between the two monomeric or polymeric compounds corresponds to the projected lengths of two methylene groups, i.e. 2.5 Å, as expected. On the other hand, the layer thickness is in agreement with the lengths of an all-trans conformation for the monomeric molecule or for the side group of the polymer with the methyl group and size of the main chain added (see figure 18). The discussion of any structural model must thus account for

$$d_{\text{monomer}} \approx d_{\text{polymer}} \approx L_{\text{molecule}}.$$

Various packing models for liquid-crystalline side group polymers have been proposed as a two layer packing model, an antiparallel one layer and an antiparallel one layer with total overlapping pack-model [7]. A parallel one layer packing model with compensation of dipole moments is also feasible. This model and the one with antiparallel one layers with total overlapping can explain the experimentally established layer thickness of smectic and crystalline phases. However, a more thorough study will be necessary for a selection of one model or the other, since the very few reflections in the X-ray patterns are insufficient for a unique interpretation, and additional information will be needed.

Figure 18. Lengths of molecular models and observed $d$-spacings in X-ray diffraction experiments.
References